

## LETTER TO THE EDITOR

# Magnetic and Structural Studies of $\text{La}_{1-x}\text{Sm}_x\text{TiO}_3$ ( $0 \leq x \leq 1$ )

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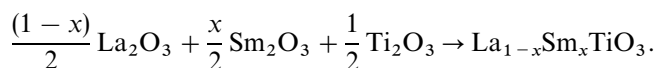
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Communicated by J. M. Honig August 25, 1997; accepted September 8, 1997

**It was found that  $\text{La}_{1-x}\text{Sm}_x\text{TiO}_3$  ( $x = 0, 0.0625, 0.1875, 0.25, 0.375, 0.5, 0.75$ , and  $1$ ) systems form a complete solid solution phase with the orthorhombic perovskite structure (GdFeO<sub>3</sub> type). Between  $x = 0$  (LaTiO<sub>3</sub>;  $T_N$  (Néel temperature)  $\sim 130$  K) and  $x = 1$  (SmTiO<sub>3</sub>;  $T_N \sim 50$  K), the susceptibility-temperature ( $\chi$ - $T$ ) curves exhibited a characteristic peak phenomenon below  $T_N$ 's only for  $0.0625 \leq x \leq 0.375$ . Such behavior almost disappears for  $x \geq 0.5$ . © 1997 Academic Press**

Orthorhombic LaTiO<sub>3</sub> is reported to be a so-called canted antiferromagnet with  $T_N$  (Néel temperature)  $\sim 140$  K (1–5). The system loses magnetic ordering and changes into a paramagnetic metal by slight oxidation of the Ti ions, caused by nonstoichiometry of the oxygen content (3) by substitution of  $\text{La}^{3+}$  for  $\text{Sr}^{2+}$  ions ( $\text{La}_{1-x}\text{Sr}_x\text{TiO}_3$ ) (3, 4), or by a slight La deficiency (5, 6). As a different approach to this system, in the present work, we prepared a solid solution of  $\text{La}_{1-x}\text{Sm}_x\text{TiO}_3$  ( $0 \leq x \leq 1$ ) to elucidate the effects of lanthanide substitution on its structural and magnetic properties. We found a characteristic peak phenomenon in the magnetization-temperature curves in some La-rich compounds. This paper briefly reports the results.

Samples were prepared by the conventional ceramic method:



The  $x$  values for  $\text{La}_{1-x}\text{Sm}_x\text{TiO}_3$  were 0.0625, 0.1875, 0.25, 0.375, 0.5, 0.75, and 1.0. Starting materials were dried La<sub>2</sub>O<sub>3</sub> (99.99%) and Sm<sub>2</sub>O<sub>3</sub> (99.99%) and as-cast Ti<sub>2</sub>O<sub>3</sub> (99.9%) (Soekawa Chemical). The reaction mixtures ( $\sim 2$  g) were thoroughly ground, pelletized, and fired in a tungsten crucible at 1550°C for 24 h in a vacuum better than  $10^{-5}$  Torr. The firing was repeated two to three times, with intermediate grinding. The samples were gray-black in color. Some of

the samples were prepared twice in a separate run, and were verified to show reproducibility in structural and magnetic properties. They were examined by the powder XRD (X-ray diffraction) method using CuK $\alpha$  radiation (Rigaku Geigerflex). The lattice parameters were calculated from least-squares fitting.

The oxygen content  $y$  in  $\text{La}_{1-x}\text{Sm}_x\text{TiO}_y$  was determined in a TGA (thermogravimetric analysis) apparatus from the weight gain of the samples when oxidized in air at 1000°C. The  $y$  value determined was  $3.02 \pm 0.02$  for each sample. For convenience, we designate the system hereafter as  $\text{La}_x\text{Sm}_{1-x}\text{TiO}_3$ .

DC magnetization measurements were made using a SQUID magnetometer (Quantum Design MPMS). Susceptibility-temperature ( $\chi$ - $T$ ) curves for each sample were measured under both the FC (field-cooled) and ZFC (zero-field-cooled) conditions. The former was measured on cooling from 300 to 4.5 K at 100 G. The latter was measured in a residual magnetization mode on heating the sample to 300 K after zero-field cooling to 4.5 K, applying a 55,000 G field, and then reducing it to 100 G. Magnetization-magnetic field ( $M$ - $H$ ) isotherm curves were measured at several temperatures with changing field between  $\pm 55,000$  G.

Lattice parameters obtained from the XRD measurements are plotted in Fig. 1 for all of the samples as a function of Sm content ( $x$ ) in  $\text{La}_{1-x}\text{Sm}_x\text{TiO}_3$ . All of the diffraction peaks originated only from the reaction products, and could be suitably assigned to the single-phase orthorhombic perovskite structure (GdFeO<sub>3</sub> type). The lattice parameters for LaTiO<sub>3</sub> ( $x = 0$ ) and SmTiO<sub>3</sub> ( $x = 1$ ) obtained here are in fair agreement with those of previous studies (2, 6–8). In Ref. (2), they are  $a = 5.601$  Å,  $b = 5.590$  Å,  $c = 7.906$  Å,  $V = 247.5$  Å<sup>3</sup> for LaTiO<sub>3</sub> and  $a = 5.660$  Å,  $b = 5.454$  Å,  $c = 7.732$  Å,  $V = 238.4$  Å<sup>3</sup> for SmTiO<sub>3</sub>. It is seen that the  $b$  and  $c$  lengths and cell volume ( $V$ ) decrease almost continuously with  $x$ , obviously owing to the lanthanide contraction. The length of the  $a$  axis increases with increasing  $x$  up to that of SmTiO<sub>3</sub> ( $x = 1$ ).

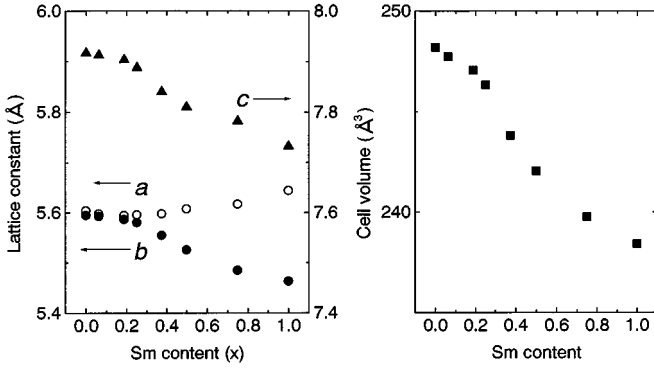


FIG. 1. Lattice parameters plotted against Sm content ( $x$ ) for  $\text{La}_{1-x}\text{Sm}_x\text{TiO}_3$ .

The composition dependence of all of the lattice parameters changes at  $x \sim 0.3$ . The changes are rather gradual for  $x < 0.3$ , but become somewhat steeper for  $x > 0.3$ . Thus, it is speculated that crystallographically the system with  $x < 0.3$  is closely similar to  $\text{LaTiO}_3$  ( $x = 0$ ) with slight modification, but tends increasingly to resemble  $\text{SmTiO}_3$  ( $x = 1$ ) with  $x > 0.3$ . The magnetization peak phenomenon was essentially distinctly observed only for  $x < 0.3$ , as discussed below.

Figure 2 shows  $\chi$ - $T$  curves for the end compounds  $\text{LaTiO}_3$  and  $\text{SmTiO}_3$  and for the solid solution  $\text{La}_{0.9375}\text{Sm}_{0.0625}\text{TiO}_3$ , where the substitution is least (6.25%) in the present work. Only the ZFC curves are shown for  $\text{LaTiO}_3$  and  $\text{SmTiO}_3$ , for they were nearly identical to the FC curves. The  $\chi$ - $T$  curve of  $\text{LaTiO}_3$  resembles those previously reported (3). This compound is called a canted antiferromagnet with  $T_N$  (Néel temperature)  $\sim 140$  K. The present  $T_N$  value is somewhat lower,  $\sim 130$  K, probably because of the slight oxidation (3). However, the magnetic moment at 4.5 K ( $\sim 9.2 \times 10^{-3} \mu_B/\text{Ti ion}$ ) is close to that in Ref. (3) ( $\sim 1.0 \times 10^{-2} \mu_B/\text{Ti ion}$ ). As far as the authors know, no detailed magnetic properties of

$\text{SmTiO}_3$  have been reported. The compound showed a transition at  $\sim 50$  K (also called  $T_N$  for convenience). The  $M$ - $H$  curve at 4.5 K indicates that the compound is nearly paramagnetic, with a small residual moment ( $\sim 7.9 \times 10^{-4} \mu_B/\text{molecule}$ ).

The  $\chi$ - $T$  curves of the mixed compound  $\text{La}_{0.9375}\text{Sm}_{0.0625}\text{TiO}_3$  are entirely different. As the temperature is lowered from 300 K in the FC run, the sample first exhibits a transition at  $\sim 110$  K. This temperature is tentatively also labeled  $T_N$ . With further decrease of the temperature, the susceptibility exhibits a characteristic broad peak at  $\sim 60$  K (peak temperature, defined as  $T_P$ ), and finally decreases at  $\sim 10$  K down to  $\sim 10^{-4}$  emu/g. In the ZFC run, a similar susceptibility peak is observed as well. However, the maximum susceptibility at  $T_P$  ( $\chi_{\max}$ ) is much less and  $T_P$  itself is quite different,  $\sim 25$  K.

A similar peak phenomenon in the  $\chi$ - $T$  curve was consistently observed for  $x \leq 0.375$ . This is seen in Fig. 3, which shows the FC  $\chi$ - $T$  curves for several representative samples. Each ZFC curve for  $x \neq 0.9375$  having the peak phenomenon exhibited a profile similar to the corresponding FC curve with much lower  $\chi_{\max}$ . The  $x = 0.375$  curve is omitted in the figure, for the features mentioned above can be hardly recognized at this ordinate scale. The values  $T_N$ ,  $T_P$ , and  $\chi_{\max}$  change with Sm content ( $x$ ). With  $x$  increasing from 0.0625 to 0.375,  $T_N$  shows a tendency to decrease from  $\sim 110$  to  $\sim 80$  K, whereas  $T_P$  increases from  $\sim 60$  to  $\sim 80$  K, accompanying the reduction in  $\chi_{\max}$ . The peak in the  $\chi$ - $T$  curve almost vanishes for  $x \geq 0.5$ , probably through the crossing of  $T_N$  and  $T_P$  around 80 K. It is therefore inferred that the slight modification of the canted spin ordering in  $\text{LaTiO}_3$  is brought about by the Sm substitution, leading to the onset of the characteristic peak phenomenon in this system. In fact, this phenomenon is almost confined to the region  $x < 0.3$ , where the change of the lattice parameters from those of  $\text{LaTiO}_3$  is rather small (Fig. 1).

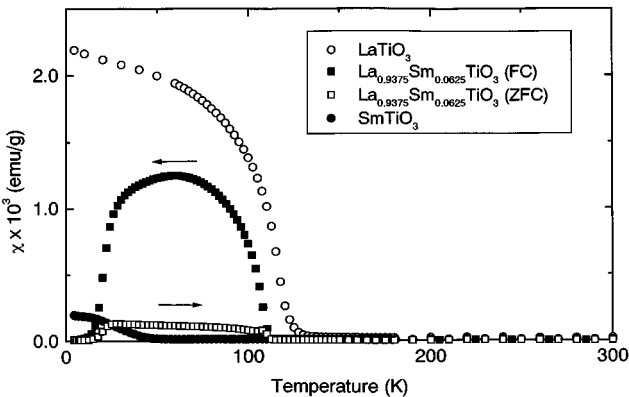


FIG. 2.  $\chi$ - $T$  curves for  $\text{LaTiO}_3$ ,  $\text{SmTiO}_3$ , and  $\text{La}_{0.9375}\text{Sm}_{0.0625}\text{TiO}_3$ .

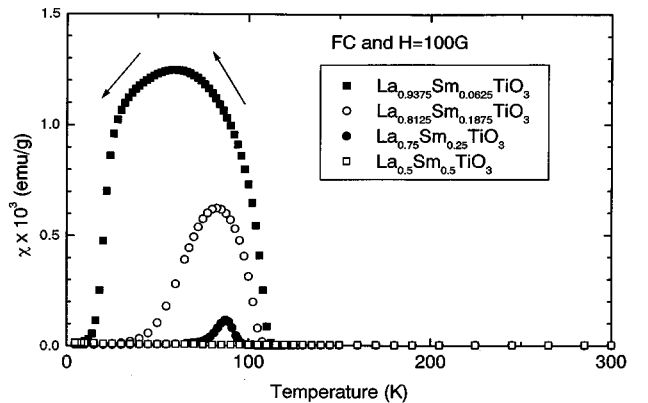


FIG. 3. The FC  $\chi$ - $T$  curves for samples other than  $\text{LaTiO}_3$  and  $\text{SmTiO}_3$ .

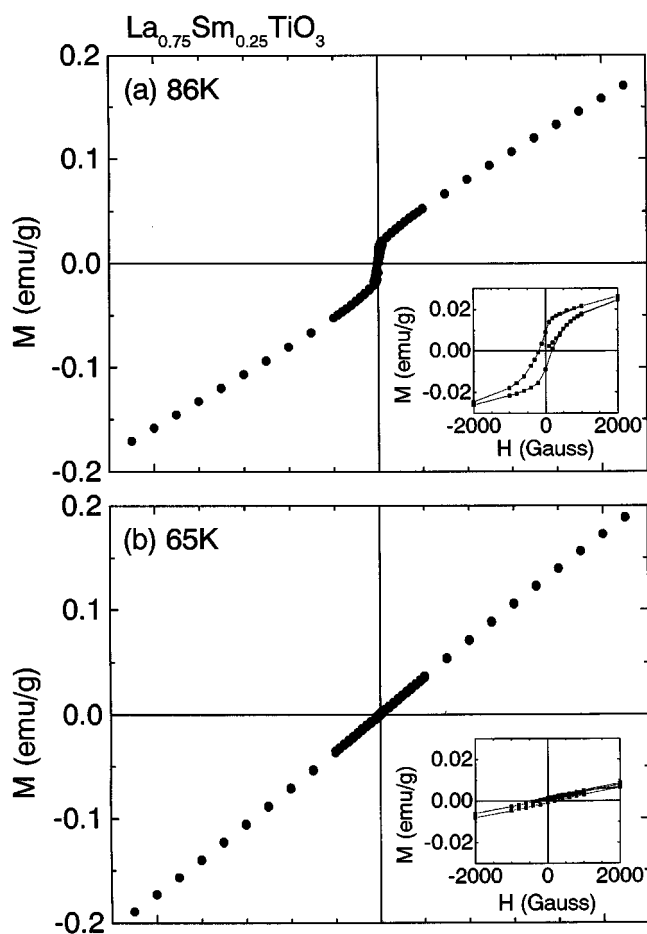


FIG. 4.  $M$ - $H$  curves measured at (a) 86 K and (b) 65 K for  $\text{La}_{0.75}\text{Sm}_{0.25}\text{TiO}_3$ .

Figure 4 shows the  $M$ - $H$  curve of  $\text{La}_{0.75}\text{Sm}_{0.25}\text{TiO}_3$  measured at 85 K ( $T_p$ ) and 65 K. Close to  $T_p$ , residual magnetization is clearly observed, whereas it is quite small at 65 K. At lower temperatures, it was found that the residual magnetization gradually increased, however, was much lower than those at  $T_p$ . For all the samples exhibiting the peak phenomenon, similar behavior of the  $M$ - $H$  curves was observed below  $T_N$ .

In summary, we found that a solid solution system  $\text{La}_x\text{Sm}_{1-x}\text{TiO}_3$  ( $x = 0, 0.0625, 0.1875, 0.25, 0.375, 0.5, 0.75$ , and 1.0) formed an orthorhombic perovskite phase. Between  $x = 0$  ( $\text{LaTiO}_3$ ;  $T_N \sim 130$  K) and  $x = 1$  ( $\text{SmTiO}_3$ ;  $T_N \sim 50$  K), a characteristic magnetization peak phenomenon was clearly observed only for  $x < 0.3$  where the crystallographic alteration from  $\text{LaTiO}_3$  is rather minor.

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